



# Porous Dendritic Platinum Nanotubes with Extremely High Activity and Stability for Oxygen Reduction Reaction

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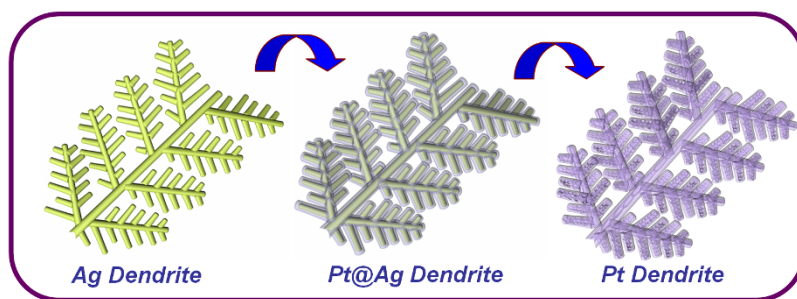
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**Controlling the morphology of Pt nanostructures can provide opportunities to greatly increase their activity and stability. Porous dendritic Pt nanotubes were successfully synthesized by a facile, cost-effective aqueous solution method at room temperature in large scale. These unique structures are porous, hollow, hierarchical, and single crystalline, which not only gives them a large surface area with high catalyst utilization, but also improves mass transport and gas diffusion. These novel Pt structures exhibited significantly improved catalytic activity (4.4 fold) for oxygen reduction reaction (ORR) and greatly enhanced durability (6.1 fold) over that of the state-of-the-art commercial Pt/C catalyst. This work provides a promising approach to the design of highly efficient next-generation electrocatalysts.**

Platinum (Pt) is the most effective catalyst for both hydrogen oxidation and oxygen reduction in proton exchange membrane fuel cells (PEMFCs)<sup>1,2</sup>. However, the high cost and insufficient stability of Pt catalysts pose a severe challenge to the commercialization of PEMFCs<sup>3</sup>. It is generally accepted that the chemical and physical properties of the metal nanostructures can be tuned by controlling their shape, size, and crystallinity. To date, many approaches have been developed to prepare Pt nanostructures in different shapes to improve their fuel cell performance: for example, multi-octahedrons<sup>4</sup>, that exhibited a higher specific activity (2.7-fold greater, while retaining similar mass activity) and stability (5.9-fold greater) than the state-of the art Pt/C catalyst; nanocubes that showed twice the specific activity of that of the commercial catalyst<sup>5</sup> and were also found to be 4 times more active than polyhedrons or truncated cubes<sup>6</sup>; nanowires that were 3 times more active and 5 times more stable than the Pt/C commercial catalyst<sup>7-9</sup>; and other shapes<sup>10,11</sup>. These results show that controlling the morphology of Pt nanostructures provides great opportunities to improve fuel cell performance. Among those materials, hollow nanostructures have attracted significant attention because their properties are distinct from those of their solid counterparts, including their large surface area, low density, rich surface chemistry, high catalytic activity, and low cost<sup>12-16</sup>. As a result of the creation of more cavities and holes and the assembly of a greater number of nanoparticles, porous hollow Pt nanostructures<sup>16-19</sup> possess a much larger surface area and more catalytically active points for electro-catalysis; therefore, they have drawn particular interest for fuel cell applications<sup>16-20</sup>. Recently, extensive interest has been expressed in the synthesis of hierarchical structures that are, ideally, connected through low dimensional materials with particularly structured junctions. This interest has arisen because such hierarchical materials not only exhibit excellent mechanical strengths and electrical conductivities that are superior to those of their constituents while maintaining comparable surface areas, but they also solve the problem of nanomaterial agglomeration<sup>21-24</sup>. There have been advances in the synthesis of hollow and porous<sup>12-20,25,26</sup> Pt nanostructures, most of which have focused on 1D Pt tubes, but such development has been rare for hierarchical porous Pt tubes. Therefore, the design of more complex Pt nanostructures by combining the multi-advantages of porous, hollow, and hierarchical structures to obtain significantly improved catalyst activity and stability are highly desired, however remains grand challenge.

In this paper, we describe a novel approach to address both the activity and stability issues synergistically by using porous dendritic Pt nanotubes as fuel cell catalysts. Such hierarchical catalysts are expected to integrate several different functionalities (hollow, dendrites, porous) in one structure, which is difficult to accomplish in a single morphology. The synthesis strategy is based on the galvanic replacement reaction between Ag dendrites and H<sub>2</sub>PtCl<sub>6</sub> that is conducted in surfactant-free, environmentally friendly aqueous solution at room temperature. Compared with the most widely used commercial catalyst, which is made of Pt nanoparticles (2–5 nm)

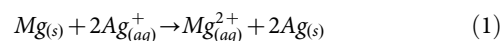


**Figure 1** | Schematic illustration of the formation of the Pt dendritic nanotubes originating from the Ag dendrite nanostructure.

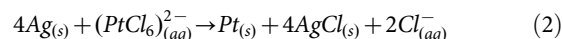
supported on carbon-particle aggregates, the micrometer-long porous dendritic Pt tubes hold great potential to significantly improve the catalytic activity and stability due to the following reasons: 1) they are self-supporting, eliminating the carbon corrosion problem that potentially causes migration of the Pt nanoparticles, coalescence, and even detachment from the catalyst system<sup>18</sup>; 2) their hierarchical structure has stronger mechanical properties and is less vulnerable to dissolution and aggregation<sup>21</sup>; 3) their porous, hollow and hierarchical structure, in addition to providing a large surface area with high catalyst utilization, can also improve mass transport and gas diffusion<sup>28,29</sup>; 4) importantly, the continuous shell of dendritic tubes is composed of numerous assembled Pt nanocrystals with the same crystal orientation, which more closely resemble to the highly active single-crystalline bulk catalyst surface of Pt<sup>29,30</sup>. These new achievements, to the best of our knowledge, have not yet been reported in the literature.

## Results

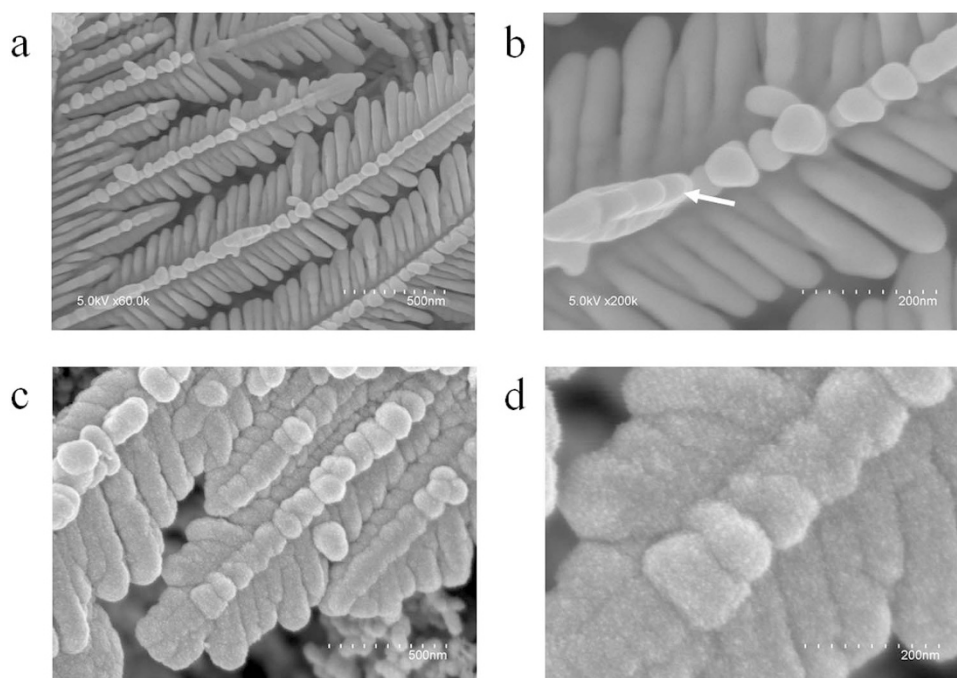
Porous Pt dendritic nanotubes were synthesized by a two-step method based on galvanic replacement reactions (Figure 1). In the first step, Ag dendrites were synthesized by the replacement reaction between commercial Mg powder and AgNO<sub>3</sub> in aqueous solution, based on the following reaction:



The morphological and structural characterization of Ag dendrites is shown in Figure S1 (Supplementary Information) and Figure 2. The low-magnification scanning electron microscope (SEM) image and the energy-dispersive X-ray spectrum (EDX) (Figure S1) indicate that micrometer long uniform Ag dendrites were successfully synthesized at a large scale. The higher magnification SEM images (Figure 2a and b) clearly show that the dendrite structures have multilevel generations with a long main trunk and many well-aligned side branches (or sub-branches, as indicated by a white arrow in Figure 2b), resembling fractal leaves. Generally, the lengths of the trunks and the (sub-)branches are several tens of micrometers and several hundreds of nanometers, respectively. However, all of the diameters of the trunks, branches, and sub-branches have a similar size of approximately 40 nm. In the second step, the as-synthesized Ag dendrites were then used as templates for the synthesis of Pt dendritic tubes through a galvanic replacement reaction, as illustrated in Figure 1, according to the following formula:



After the completion of the reaction, the SEM image (Figure 2c) reveals that similar dendritic structures were formed but with



**Figure 2** | SEM characterization of Ag dendrites and porous dendritic Pt tubes. SEM images (a, b) of Ag dendrites and (c, d) of porous dendritic Pt nanotubes, at different magnifications. Platinum exactly replicates the dendritic structure of the Ag with relatively larger branches and a rough surface.

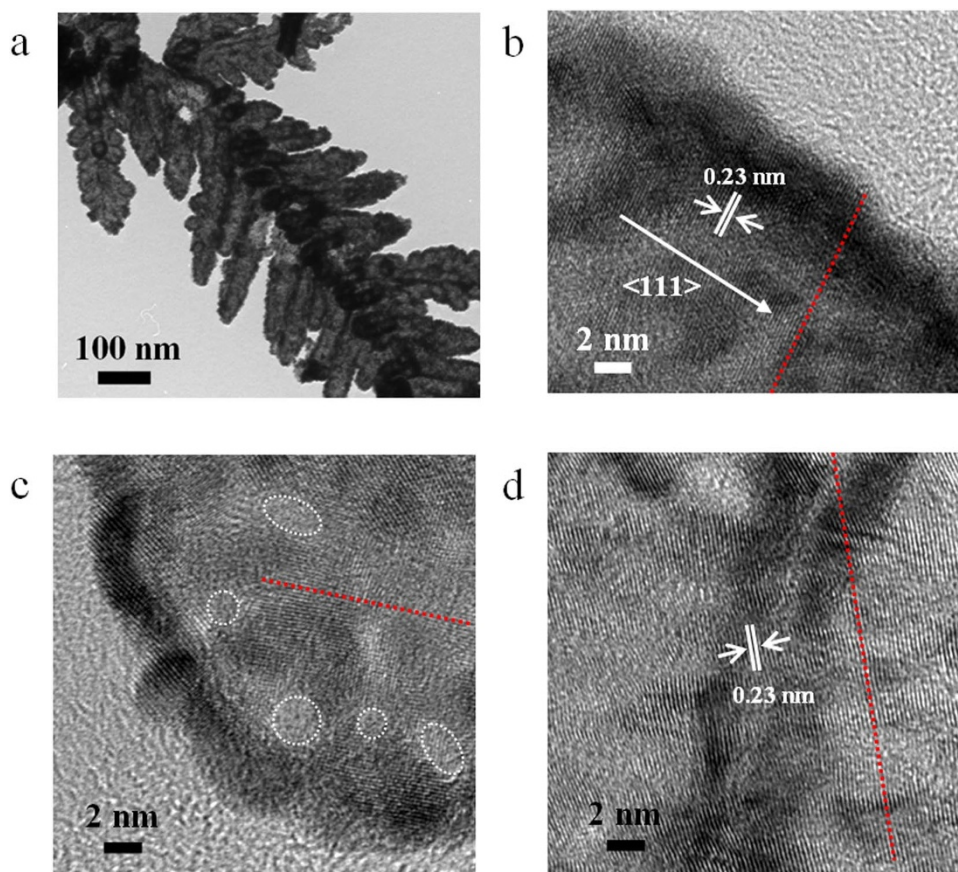


relatively greater branch diameters (ca. 46 nm). This is consistent with the so-called Kirkendall effect that has been used to illustrate the outward growth of hollow structures<sup>31–34</sup>. The surface of dendrites becomes rough and develops a porous structure that is composed of numerous assembled tiny nanoparticles (Figure 2d). The EDX spectrum, shown in Figure S2, confirms the formation of pure platinum. Significantly, the amount of product can be easily scaled up to gram scale (e.g. 10 g Ag dendrites and 1.5 g Pt dendritic tubes, Figure S3) by simply multiplying the amounts of reactants, which is important for the commercialization of PEM fuel cells.

The Pt products were further characterized by transmission electron microscope (TEM). It is found that the centers of the dendritic Pt nanostructures are brighter than the edges, clearly depicting a tubular morphology (Figure 3a), and the wall thickness of the Pt nanotube branches is  $\sim 2.5$  nm (Fig. 3b–d), which is slightly larger than the theoretical calculated value of 2.0 nm if the wall is assumed to be fully condensed (see the calculation shown in Supplementary Information Figure S4). The high-resolution TEM (HRTEM) image (Fig. 3b) shows the representative morphology of an individual branch of Pt dendrites, which indicates that the shells of the nanotubes consist of tiny Pt nanocrystals. Interestingly, these Pt nanocrystals assemble to form a continuous shell with single-crystal structure (as indicated by the red dot line). Due to the large diameter of the tubes (about 46 nm, with low surface curvature), the shell more closely resembles to a bulk single-crystal surface of Pt that has an extremely high catalytic activity<sup>29,30</sup>. The lattice spacing along the axis of the nanotube branch is ca. 0.23 nm, corresponding to the interplanar spacing of {111} planes in the fcc Pt. The HRTEM image shown in Figure 3c was taken from a thin part of a dendritic branch. This image clearly shows a more porous structure in the thinner part

of the shell, as indicated by the white dotted circles, and most of the small nanocrystals aligned with the same crystal orientation (as indicated by the red dot line). Figure 3d was taken from the boundary between two Pt nanotube branches; interestingly, different tubes have the same crystal orientation (as indicated by the red dot line). This phenomenon may be caused by the epitaxial growth of the Pt that was templated from the single crystalline Ag dendrite. The detailed Ag structural characterization, including the crystallinity and the (sub)branch growth directions, are presented in Figure S5, which demonstrates that the Ag dendritic single crystal grows along a preferential direction of  $\langle 111 \rangle$ . The small mismatch in the lattice constants between fcc Ag (4.08 Å) and fcc Pt (3.92 Å) favors the epitaxial growth of Pt on Ag dendrites, therefore forming continuous Pt shells. In addition, it should be noted that these TEM images were obtained after the removal of unreacted Ag template by acid wash and the strong sonication during TEM sample preparation process. The Pt shells remain uncollapsed as hollow nanotubes, which suggests the excellent mechanical stability of the dendritic hollow structures.

To study the transformation process from Ag dendrites to Pt dendritic nanotubes, samples were taken from the solution as a function of reaction time, and the morphology and composition of these samples were analyzed by TEM and powder X-ray diffraction (XRD). The TEM images show different stages of the galvanic replacement reaction between Ag dendrites and  $\text{H}_2\text{PtCl}_6$  (Figure S6). After 1 min of reaction, some Pt nanoparticles began to appear on the outer surface of the Ag dendrites, and some parts of Ag dendrites started to get porous and hollow (as indicated by arrows in Figure S6b). This was because that four Ag atoms were consumed to form only one Pt atom, as defined by the stoichiometric relationship between Ag and



**Figure 3** | TEM characterization of porous dendritic Pt tubes. (a) TEM images of the dendritic Pt tubes. (b, c) HRTEM images taken from different areas of the Pt tubes with relatively thick or thin wall thicknesses, respectively. (d) HRTEM image of the boundary between two Pt nanotubes. Dendritic Pt nanotubes exhibited a well-aligned crystalline nature.





$\text{H}_2\text{PtCl}_6$  (Eq. (2)), leaving behind vacancies in the template as the galvanic replacement reaction proceeds. After 8 min and 12 min of reaction, the whole dendrites get more transparent than the solid Ag templates, indicating the formation of porous and hollow Pt tubes (Figure S6c,d). In addition, their surface become rougher (as indicated by the arrows) than the smooth surface of original Ag dendritic template (Figure S6a), indicating that Pt gradually increased and evolved into nanoshells around the Ag templates. Finally, high quality dendritic porous Pt nanotubes with uniform sizes were obtained as a result of complete consumption of Ag templates after 24 hours.

Further, the corresponding XRD patterns (Figure 4) of the above samples also indicate the evolution stages from Ag to pure Pt. Figure 4a shows the purity of the Ag dendrites (JCPDS, 04-0783), with peaks located at 38.12, 44.27 and 64.43 that can be assigned to the (111), (200) and (220) planes of Ag, respectively. After an 8 min reaction, the Ag peaks are dramatically reduced, and obvious AgCl peaks appear (Figure 4b). When the reaction proceeded for 12 min, the peak ratio of Ag to AgCl decreased. The Pt peaks were not obvious, mainly because the atomic ratio of Pt to AgCl in the product is 1:4 (according to the formula  $(\text{PtCl}_6)^{2-} + 4\text{Ag} \rightarrow \text{Pt} + 4\text{AgCl} + 2\text{Cl}^-$ ), and Pt is an element with heavy atoms. Therefore, the XRD pattern shows a strong AgCl signal (Figure 4c). When the samples were washed using saturated NaCl and  $\text{HNO}_3$  to eliminate the AgCl and Ag, respectively, pure Pt was obtained, as shown in Figure 4d. When making a careful comparison with the XRD patterns of the 8 and 12 min products, the contribution of the Pt (Figure S7) is visible. In fact, the peaks located at 38.8 and 45.0 are composite peaks, composed of two peaks at 38.12 (Ag 111) and 39.76 (Pt 111), as well as 44.28 (Ag 200) and 46.24 (Pt 200), respectively. The intensities of these two peaks decrease, and the corresponding peak positions shift positively with reaction time, indicating the gradual sacrifice of Ag and formation of Pt.

The electrochemical properties of the porous Pt dendritic tubes were evaluated, and the results were compared with those obtained for a commercial Pt/C catalyst (E-TEK, 30 wt% Pt, made of Pt nanoparticles on a carbon support) and Pt black. Figure 5a shows the cyclic voltammetry (CV) curves of these three catalysts recorded at room temperature in  $\text{N}_2$  purged 0.5 M  $\text{H}_2\text{SO}_4$  solution with a sweep rate of 50 mV/s. All three CV curves exhibited strong peaks associated with hydrogen adsorption/desorption below 0.4 V (vs. RHE) and Pt oxide formation/reduction above that value. The

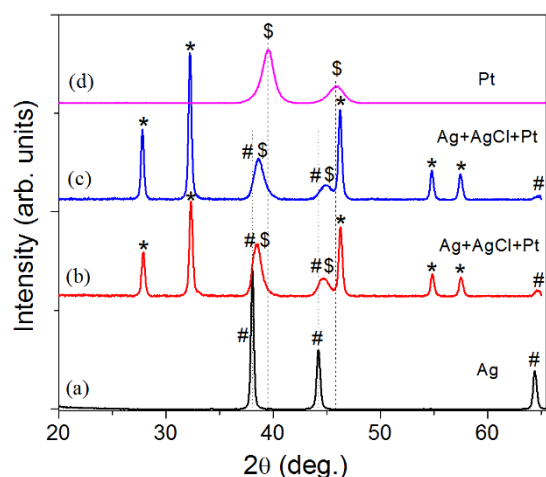
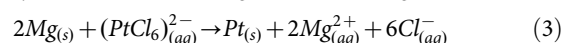
electrochemical active surface areas (ECSA) were calculated by measuring the charge ( $Q_{\text{H}}$ ) associated with the adsorbed hydrogen monolayer ( $\text{H}_{\text{ads}}$ ,  $\text{H}^+ + \text{e}^- = \text{H}_{\text{ads}}$ , where  $\text{H}_{\text{ads}}$  refers to the underpotentially deposited hydrogen) between  $0 < E < 0.4$  V (where E is the applied potential) and assuming that  $Q_{\text{ref}} = 0.21$  mC/cm<sup>2</sup> for the adsorption of a hydrogen monolayer<sup>43,44</sup>, corresponding to a surface density of  $1.3 \times 10^{15}$  Pt atoms per cm<sup>2</sup>. The specific ECSA of the Pt dendritic tubes was 23.3 m<sup>2</sup>/g Pt, which is 1.4 times greater than that of Pt black (16.3 m<sup>2</sup>/g Pt), but is around half (53%) that of commercial Pt/C catalyst (43.8 m<sup>2</sup>/g Pt). Interestingly, a careful inspection of the voltammograms in Figure 5a reveals that the morphological change of the Pt nanostructures from solid particles (0D) to dendritic hollow tubes (3D) also greatly altered their ability to adsorb hydroxyl species ( $\text{OH}_{\text{ad}}$ ), which has an inhibiting (site blocking) effect on the ORR activity. There are significantly positive shifts, 71 mV and 95 mV, respectively, in hydroxyl adsorption/desorption on Pt dendritic nanotubes than that on Pt/C and Pt black catalysts, indicating a slower hydroxyl adsorption rate and faster hydroxyl desorption rate on the Pt dendritic tube surface, and better ORR activity as a result<sup>45</sup>. Moreover, in agreement with the onset potential of adsorption, on Pt dendritic nanotubes, the fractional coverage by  $\text{OH}_{\text{ad}}$  ( $\Theta_{\text{OH}_{\text{ad}}}$ ) was dramatically reduced at all potential range  $>0.6$  V compared to that of Pt/C and Pt black (Figure 5b). The much lower  $\text{OH}_{\text{ad}}$  coverage on the Pt dendritic nanotubes than on the Pt/C and Pt black catalysts contributes to enhanced ORR activity and stability<sup>46,47</sup>.

Figure 5c shows the ORR polarization curves for porous Pt dendritic tubes, Pt black and Pt/C (E-TEK) catalysts obtained at room temperature in  $\text{O}_2$ -saturated 0.5 M  $\text{H}_2\text{SO}_4$  with a GC rotating disk electrode (RDE) at 1600 rpm. The half-wave potential for the porous Pt dendritic tubes shows an obvious positive shift relative to those for both the Pt black and Pt/C catalysts, indicating a higher catalytic activity for the Pt dendritic tubes than those of the other two commercial Pt nanoparticle-based catalysts. Both the mass activity and specific activity are good indicators of the quality of an electrocatalyst. As shown in Figure 5d, the Pt dendritic tubes exhibited a mass activity of 0.21 mA/ $\mu\text{g}$  Pt at 0.9 V (vs. RHE), which is 2.33 times that of the Pt/C (E-TEK) catalyst (0.09 mA/ $\mu\text{g}$  Pt) and 5.25 times that of the Pt black catalyst (0.04 mA/ $\mu\text{g}$  Pt). Interestingly, this improvement occurred in spite of the approximately 50% lower Pt active surface area of the Pt dendritic tube catalyst. On the basis of both the mass activity and the active surface area, a specific ORR activity of 0.901 mA/cm<sup>2</sup> Pt for the porous Pt dendritic tubes was obtained at 0.9 V, which is higher than the US department of energy (DOE) 2015 target for the specific activity (0.720 mA/cm<sup>2</sup> Pt)<sup>48</sup>, and it is 4.40-fold and 3.68-fold greater than those of the E-TEK Pt/C catalyst (0.205 mA/cm<sup>2</sup> Pt) and the Pt black catalyst (0.245 mA/cm<sup>2</sup> Pt), respectively. This improvement agrees well with those reported for Pt tubes obtained under full-cell operating conditions<sup>7</sup>.

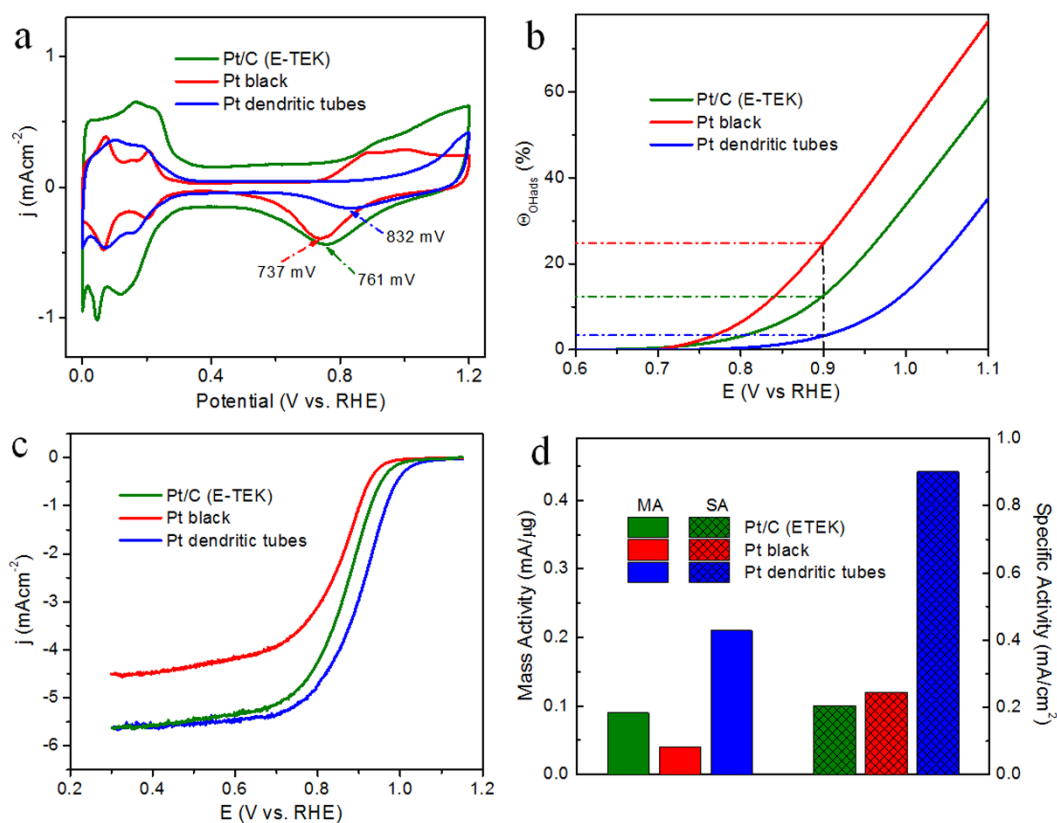
The accelerated durability tests (ADT) of the catalysts were conducted by applying linear potential sweeps between 0.6 and 1.2 V (vs. RHE) in  $\text{O}_2$ -saturated 0.5 M  $\text{H}_2\text{SO}_4$  solution at room temperature with a scan rate of 50 mV/s. After 4,000 cycles, the CV measurements showed that the Pt dendritic tube catalyst lost only 11% of the Pt initial ECSA (Figure 6a,d), whereas the degradations of both Pt nanoparticle-based commercial catalysts were quite severe, with 67.5% and 52.7% losses of the initial Pt ECSA for E-TEK Pt/C (Figure 6b,d) and Pt black (Figure 6c,d), respectively. These results indicate that the porous Pt dendritic tube catalyst is 6.1-fold and 4.8-fold more stable than the Pt/C and Pt black nanoparticle-based commercial catalyst, respectively.

## Discussion

The similar replacement reaction was extended between Mg and  $\text{H}_2\text{PtCl}_6$  to synthesize Pt, according to the following equation:



**Figure 4 | XRD characterization.** XRD patterns showing the stages of evolution from pure Ag to pure Pt, as a function of reaction time for (a) 0 min, (b) 8 min, (c) 12 min, and (d) final product. The pound sign (#) indicates the silver diffraction peaks; the asterisk sign (\*) indicates the silver chloride diffraction peaks; and the dollar sign (\$) indicates the platinum peaks.

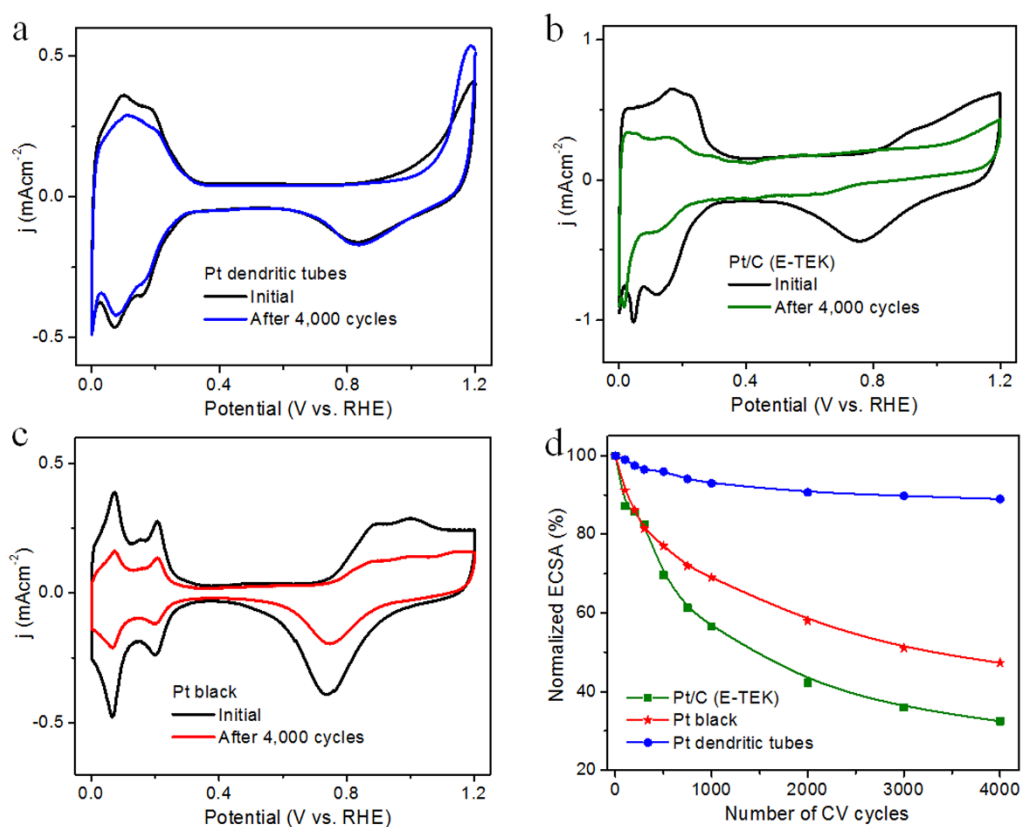


**Figure 5 | Electrochemical evaluation.** (a) CV curves and (b) hydroxyl surface coverage ( $\Theta_{OH}$ ) for three types of catalysts: Pt dendritic tubes (blue line), Pt black (red line) and Pt/C (E-TEK, green). The much lower  $OH_{ad}$  coverage on the dendritic Pt nanotubes than on the Pt/C and Pt black catalysts could contribute to the enhanced ORR activity. (c) Polarization curves for ORR of the Pt dendritic tube (blue), Pt black (red line) and Pt/C (E-TEK, green, line) catalysts in  $O_2$ -saturated 0.5 M  $H_2SO_4$  solution at room temperature (1600 rpm, sweep rate 10 mV/s). (d) Mass activity and specific activity at 0.9 V (vs. RHE) for the three catalysts. At 0.9 V (vs. RHE), Pt dendritic tubes were 2.33 times and 5.25 times more active, on the basis of equivalent Pt mass, than Pt/C and Pt black catalysts, respectively. Furthermore, the porous Pt dendritic tubes exhibited a higher specific ORR activity (0.901 mA/cm<sup>2</sup> Pt) than the US DOE 2015 target (0.720 mA/cm<sup>2</sup> Pt).

The original Mg particles are irregular in shape and quite large in size (20–100  $\mu\text{m}$ ), and their surface are rough with some cracks (Figure S8a,b). After the reaction, urchin-like structures that are composed of many Pt nanoparticles were obtained (Figure S8c,d), which is different to the morphology of either original Mg particles (Figure S8a,b) or Ag dendrites (Figure 2a,b, obtained by Mg and  $AgNO_3$ ). Previously, we have given a detailed observation on morphology evolution and explanation on the formation of Ag dendrites obtained from Mg and  $AgNO_3$ <sup>35</sup>. It is worthwhile to clearly understand why so much different structures, including urchins, dendrites, and dendritic tubes, were obtained via similar replacement reactions involving different precursors (Mg and  $H_2PtCl_6$ , Mg and  $AgNO_3$ , as well as Ag and  $H_2PtCl_6$ ). First, the driving force for the galvanic replacement reaction is the electrical potential difference between two metals. The net potentials in the reactions of Mg and  $H_2PtCl_6$  (3.40 V) as well as Mg and  $AgNO_3$  (3.16 V) are much greater than that in the reaction of Ag and  $H_2PtCl_6$  (0.24 V) [ $Mg^{2+}/Mg$  (-2.36 V),  $Ag^+/Ag$  (0.80 V), and  $Pt^{4+}/Pt$  (1.04 V) vs. the standard hydrogen electrode (SHE)]<sup>36,37</sup>. Accordingly, the reaction rates of the former two are much faster than the latter. In addition, because the difference in crystal structures and lattice constants, Mg (hcp;  $a = 3.21 \text{ \AA}$ ,  $c = 5.21 \text{ \AA}$ ) and Ag (fcc;  $a = 4.08 \text{ \AA}$ )/Pt (fcc;  $a = 3.92 \text{ \AA}$ ) the epitaxial growth of Ag (or Pt) on Mg particle templates is not favorable. Therefore, the products of the former two reactions do not follow the morphology of their original templates (the sacrificial metal, Mg). Further, comparing the former two reactions (Mg +  $H_2PtCl_6$  and Mg +  $AgNO_3$ ), their net potentials (3.40 V vs. 3.16 V) are very close, and thus the reaction rates are quite similar.

However, based on the stoichiometric relationship defined in Eqs. (1) and (3), the molar ratio of Ag to Mg is 2:1, while the molar ratio of Pt to Mg is 1:2. That means two Mg atoms could produce, *in situ*, four Ag atoms, while only one Pt atom, resulting an expanding growth for Ag while a shrinking growth for Pt. Therefore, larger size and solid structure for Ag was obtained while smaller structure composed of nanoparticles were formed for Pt. In the case of Pt dendritic tubes, first, the net potential for Eq. (2) is very low (0.24 V), thus the reaction rate is much slower; second, the fcc lattice constants of Ag and Pt are very close. These two factors allow i) the continuous Pt deposition on the surface of Ag template replicating the dendritic structure, and ii) the reconstruction of Pt shells into highly crystalline structure via processes such as Ostwald ripening<sup>38</sup>.

Based on the above analysis, a possible mechanism for the formation of dendritic Pt tubes is proposed in Figure S9. To simplify, a single branch of dendritic nanotubes is taken as a model for the illustration. As it has been established in previous studies by others and ourselves<sup>39–42</sup>, the replacement reaction starts from the highest energy sites, usually those with steps or point defects or stacking faults on the templates. Our previous work has shown the existence of defects on Ag dendrites<sup>35</sup>. Once reaction happens on such defects, they will serve as active sites for further replacement reaction (Figure S9b). The Ag atoms in the interior will diffuse to these reaction sites and react with  $H_2PtCl_6$  to generate Pt atoms. The Pt atoms prefer to nucleate and grow on the surface of the template because Pt and Ag have the same fcc structure and essentially identical lattice constant, favoring epitaxial growth of Pt on Ag dendrites. According to the stoichiometric relationship of Ag and  $H_2PtCl_6$ , four Ag atoms are



**Figure 6 | Electrochemical durability evaluation.** CV curves for (a) supportless Pt dendritic tubes and (b) Pt/C (E-TEK) and (c) Pt black catalysts before (black curves) and after 4,000 cycle ADT tests (solid curves in blue, green and red, respectively). d) Loss of electrochemical surface area (ECSA) of Pt dendritic tubes (blue line), Pt black (red line) and Pt/C (E-TEK, green line) catalysts as a function of cycling numbers in O<sub>2</sub>-purged 0.5 M H<sub>2</sub>SO<sub>4</sub> solution at room temperature (0.6–1.2 V vs. RHE, sweep rate 50 mV/s). The porous Pt dendritic nanotubes were 6.1-fold and 4.8-fold more stable than Pt/C and Pt black nanoparticle-based commercial catalysts, respectively.

consumed to form only one Pt atom, leaving behind a huge number of vacancies in the template as the replacement reaction proceeds (Figure S9c). These vacancies will promote the migration of Ag and Pt atoms significantly, to form tube-like structure. As the reaction proceeds, the concentration of the Pt<sup>4+</sup> will decrease significantly, which slows down the replacement reaction, providing the time for the atom diffusion and coalescence via Oswald ripening, to counter the increase of the surface energy. Through enough time for the reorganization of Pt atoms, uniform Pt dendritic tubes with continuous single crystalline Pt shells are formed. After thorough wash of the sample to remove AgCl and the left of Ag, pure Pt dendritic tubes are finally obtained (Figure S9d).

The highly improved activity of the Pt dendritic tubes, compared with commercial catalyst (Pt nanoparticles on carbon black), can be attributed to the following reasons: (1) the changes in the geometry and morphology (3D vs. 0D) because the 3D shape facilitates the reaction kinetics and electron transport and also improves the diffusion of O<sub>2</sub> to Pt surface; (2) the porous and hollow structure, which not only provides a large surface area for high catalyst utilization, but also improves the mass transport and gas diffusion<sup>27,28</sup>; and (3) fewer surface defects borne by the dendritic tubes, which bear a closer resemblance to the smooth single-crystalline bulk catalyst surface that have been proved to exhibit much higher ORR specific activities, for example, Pt (111) surface is 9 times more active than the current state-of-the-art Pt/C catalysts in PEMFCs<sup>29</sup>. The ORR is a multi-electron reaction ( $O_2 + 4H^+ + 4e^- = 2H_2O$ ) that may include a number of elementary steps involving different reaction intermediates<sup>49</sup>. Some charged oxygen atoms react with protons to create hydroxide molecules ( $2H_2O = OH_{ad} + H_3O^+ + e^-$ , where OH<sub>ad</sub> is the adsorbed hydroxyl layer) and likely other oxides as well. These

oxides have an affinity for Pt atoms, binding to their surface, and block access to as many as 45% of the Pt atoms<sup>46</sup>. Given that  $\Theta_{OH_{ad}}$  is greatly weakened on Pt dendritic tube surface (as show in Figure 5a,b), the key factor that determines the excellent catalytic activity of Pt dendritic tubes is the low coverage by OH<sub>ad</sub>. That means there are much fewer OH<sub>ad</sub> bind to Pt surface, leaving much more free Pt sites open to carry out oxygen reduction reactions<sup>50,51</sup>.

In order to further demonstrate the excellent electrochemical stability of Pt dendritic tubes, the changes of their morphology were investigated by TEM before and after ADT cycling (Figure 3 and Figure S10). The overall structure of the dendritic tubes retained largely intact, although the surface topology has slight change with occasionally broken in some parts (Figure S10a). HRTEM study indicates that the surface of these Pt tubes became smooth and the population of surface bump diminished after the multiple cycles, while most of the Pt primary crystals of the continues shell still keep the same orientation (Figure S10b). The significant improvement of the stability for dendritic Pt tubes is mainly attributed to the following reasons: 1) the less hydroxyl coverage (i.e. less Pt oxides formation) on large single crystal surface (large radii of curvature) of dendritic nanotubes pulls less Pt atoms off from the catalyst surface, drastically enhancing the catalyst stability; 2) the hierarchical structures, with strong mechanical property, decrease the detachment (drop-off), dissolution and aggregation of Pt from the dendritic matrix; 3) the larger radii of curvature of Pt shells make them more resistant to surface area loss via mechanisms such as Pt dissolution and re-deposition<sup>30</sup>; and 4) the elimination of corrosion of the catalyst support (e.g., carbon) due to the capability of self-supporting of micrometre-long Pt dendritic tubes.





In conclusion, porous Pt dendritic nanotubes have been successfully synthesized by a simple, cost-effective, aqueous solution method based on the galvanic replacement reaction between Ag dendrites and  $\text{H}_2\text{PtCl}_6$ . The synthesis is green and environmentally benign, as it does not require high temperature, organic solvents, or electrochemical deposition. The porous Pt dendritic nanotubes were 2.33-fold more active on the basis of equivalent Pt mass, 4.4-fold more active in terms of specific activity for ORR than the state-of-the-art Pt/C commercial catalyst, 5.25-fold more active on the basis of equivalent Pt mass and 3.68-fold more active in terms of their specific activity than the first-generation supportless Pt black catalyst. Moreover, the porous Pt dendritic nanotubes were 6.1-fold and 4.8-fold more stable than the Pt/C and Pt black catalysts, respectively. Further enhancement of the ORR activity and stability is expected through the optimization of the morphology (shape and wall-thickness) and composition (alloys) of the porous dendritic nanotubes by varying the experimental parameters and adding other elements (e.g., Au and Ni) to form alloys. Our work provides a promising approach for the design of highly active and stable next-generation catalysts with a substantial reduction in Pt loading without compromising performance. Porous Pt dendritic nanotubes may also be useful in catalysis applications beyond fuel cells.

## Methods

**Chemicals.** Doubly distilled deionized (DI) water was used for all preparations. Silver nitrate ( $\text{AgNO}_3$ , 99.95%) and hexachloroplatinic acid ( $\text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$ , 99.95%) were obtained from Sigma-Aldrich and used as received. The magnesium metal powder (Mg,  $-100 + 200$  mesh, 99.6%) was obtained from Alfa Aesar. Commercial Pt/C (E-TEK, 30 wt% Pt nanoparticles on Vulcan XC-72 carbon) and Pt black (fuel cell grade) catalysts were obtained from Sigma-Aldrich.

**Synthesis.** To synthesize Ag dendrites, a controlled amount of freshly prepared  $\text{AgNO}_3$  aqueous solution was quickly added to a glass vial that contained a predetermined amount of commercially available Mg powder. For example, the replacement reaction between Mg (5.3 mg) and  $\text{AgNO}_3$  solution (0.03 M, 17 ml) was used, in which the atomic concentration of  $\text{Ag}^+$  was over 2-fold greater than that of Mg, so that the Mg would be completely oxidized to  $\text{Mg}^{2+}$  and precipitate only Ag dendrites. The reactions were conducted in glass vials (20 ml) at  $12^\circ\text{C}$  in ambient atmosphere for several minutes. The temperature was controlled by a constant temperature bath (VWR model 1162A). After the reduction reactions were completed, the products were washed several times in DI water, collected by filtration and then dried in a vacuum oven overnight. The synthesis of Pt dendritic tubes was conducted at room temperature via the replacement reaction between the Ag dendrites (47 mg) and the  $\text{H}_2\text{PtCl}_6$  solution (0.06 M, 2.8 ml). The solution color gradually changed from golden to nearly colorless, indicating the completion of the reaction; then, the Pt dendritic tubes were thoroughly washed using saturated NaCl and  $\text{HNO}_3$  to remove AgCl and Ag, respectively, generating pure product.

**Characterization.** The morphologies, crystallinities, and chemical compositions of the samples were determined by a field-emission scanning electron microscope (FESEM, Hitachi S-4800, operating at 5 kV), which was equipped with an energy-dispersive X-ray spectrometer (EDX), a transmission electron microscope (TEM, Hitachi 7000, operating at 100 kV), and a high-resolution TEM (HRTEM, JEOL 2010F, operating at 200 kV). Powder X-ray diffraction (XRD) measurements were performed on a Bruker D8 Discover diffractometer operating at 40 kV and 40 mA with  $\text{Cu K}\alpha$  radiation ( $\lambda = 0.154$  nm).

**Electrochemical measurements.** Prior to the deposition of the catalyst onto the electrode surface for electrocatalytic assessment, a glassy carbon (GC) electrode (5 mm diameter) was polished with  $\text{Al}_2\text{O}_3$  paste (0.05  $\mu\text{m}$ ) and washed ultrasonically in double-distilled water. To prepare the catalyst suspension, the Pt dendritic nanotubes were first dispersed in a 5 mL aqueous solution containing 1 mL of isopropanol and 30  $\mu\text{L}$  of 5 wt% Nafion solution ( $V_{\text{water}} : V_{\text{isopropanol}} : V_{\text{Nafion}} = 4/1/0.0017$  in volume ratio) via sonication for 12 min. Then, 10  $\mu\text{L}$  of the catalyst suspension was cast onto the clean GC electrode and dried under flowing  $\text{N}_2$  at room temperature, leading to a Pt loading of 7  $\mu\text{g}_{\text{Pt}}$ . The electrochemical properties were measured on an Autolab potentiostat/galvanostat (Model, PGSTAT-30, Ecochemie, Brinkman Instruments) with rotation control (MSR, Pine Instruments) using a three-electrode system. The GC rotating disk electrode (RDE) on which the catalyst was loaded was used as working electrode. A Pt wire and Ag/AgCl (3 M NaCl) were used as the counter and reference electrodes, respectively. For convenience, all potentials in this study are referenced to the reversible hydrogen electrode (RHE). The solutions were de-aerated by bubbling ultra-high-purity  $\text{N}_2$  for 20 min and protected with a nitrogen atmosphere during the entire experimental procedure. All electrochemical experiments were carried out at room temperature.

For comparison, the commercial Pt/C (30 wt% Pt, E-TEK) and Pt black catalyst suspensions were prepared following a similar protocol (i.e., in a mixture of distilled water, isopropanol, and 5 wt% Nafion solution,  $V_{\text{water}} : V_{\text{isopropanol}} : V_{\text{Nafion}} = 4 : 1 : 0.0017$ ) and sonicated for 12 min. An aliquot of the catalyst suspension was pipetted onto the GC electrode, leading to a Pt loading of 7  $\mu\text{g}_{\text{Pt}}$  for all catalysts. The catalyst films were dried under flowing  $\text{N}_2$  at room temperature. All of the testing conditions and parameters were the same as those used for the Pt dendritic nanotubes.

Before collecting the effective cyclic voltammetry (CV) measurements, the working electrode was first cycled between 0 and 1.2 V at 50 mV/s for 50 cycles in an  $\text{N}_2$  purged 0.5 M  $\text{H}_2\text{SO}_4$  solution at room temperature, to produce a clean electrode surface. The electrochemical surface areas (ECSA) were calculated by measuring the charge associated with the  $\text{H}_{\text{ads}}$  ( $Q_{\text{H}}$ ) between 0 and 0.4 V and assuming that  $Q_{\text{ref}} = 0.21$  mC/cm<sup>2</sup>, corresponding to a surface density of  $1.3 \times 10^{15}$  Pt atoms per cm<sup>2</sup>, which is generally accepted for polycrystalline Pt electrodes. The ECSA of the Pt was calculated based on the following relationship:

$$\text{ECSA} = Q_{\text{H}} / (Q_{\text{ref}} \times m)$$

where  $Q_{\text{H}}$  is the charge for H adsorption (mC/cm<sup>2</sup>),  $m$  is the Pt loading (mg/cm<sup>2</sup>) in the electrode, and  $Q_{\text{ref}}$  is the charge required for the adsorption of a hydrogen monolayer on a Pt surface (0.21 mC/cm<sup>2</sup>)<sup>43</sup>. The adsorption of hydroxyl species was calculated based on the  $\text{OH}_{\text{ad}}$  peak in the CV curves at a potential higher than 0.6 V (vs. RHE). Dividing the hydroxyl adsorption area by the overall active surface area produced the surface coverage of the  $\text{OH}_{\text{ad}}$  species ( $\Theta_{\text{OHad}}$ )<sup>45</sup>. The oxygen reduction reaction (ORR) experiments were performed in an  $\text{O}_2$ -saturated 0.5 M  $\text{H}_2\text{SO}_4$  solution at room temperature with an RDE rotating rate of 1,600 rpm and a sweep rate of 10 mV/s. The current densities were normalized to the geometric area of the GC RDE (0.196 cm<sup>2</sup>). The CV measurements for the accelerated durability test (ADT) were conducted by potential cycling between 0.6 and 1.2 V (vs. RHE) for 4,000 cycles in  $\text{O}_2$ -saturated 0.5 M  $\text{H}_2\text{SO}_4$  at room temperature at a scan rate of 50 mV/s. The full-scale voltammograms between 0 and 1.2 V in  $\text{N}_2$ -saturated 0.5 M  $\text{H}_2\text{SO}_4$  were recorded periodically to track the degradation of the Pt catalysts.

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## Author contributions

G.Z. and S.S. performed the catalyst preparation, characterizations and performance tests. G.Z. and R.L. contributed to TEM characterization. Y.Z. contributed to schematic design. S.S. and G.Z. designed the study, analyzed the data and co-wrote the paper. X.S. and M.C. proposed, planned and supervised the project. All the authors participated in the discussion of the research and commented on the manuscript.

## Additional information

Supplementary information accompanies this paper at <http://www.nature.com/scientificreports>

**Competing financial interests:** The authors declare no competing financial interests.

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